Cite this: Chem. Sci., 2011, 2, 1173

www.rsc.org/chemicalscience

EDGE ARTICLE

Chemical tuning of CO₂ sorption in robust nanoporous organic polymers[†]

Robert Dawson, Dave J. Adams and Andrew I. Cooper*

Received 17th February 2011, Accepted 9th March 2011 DOI: 10.1039/c1sc00100k

We report here the carbon dioxide sorption properties for a series of conjugated microporous polymer (CMP) networks. These CMP materials incorporate a range of chemical functionalities including carboxylic acids, amines, hydroxyl groups, and methyl groups. The carboxylic acid functionalised network, rather than its amine analogue, shows the highest isosteric heat of sorption for CO_2 . This supports recent computational predictions for metal–organic frameworks and suggests that acid-functionalised frameworks could outperform more widely studied amine sorbents in CO_2 capture and separation application.

Introduction

The capture of carbon dioxide produced via the burning of fossil fuels is an important challenge for society.^{1,2} Carbon dioxide capture in porous adsorbents is one possible approach. The ideal properties for a CO₂ adsorbent will depend on the intended point of use. In pre-combustion capture of CO_2 from syngas (H₂, CO_2) and CO), or from natural gas reserves, the gas stream contains approximately 35.5% CO2 and requires materials to be used at relatively high temperatures and pressures. For post-combustion capture, the CO_2 percentage in the gas stream is lower (15–16%) and there is also a higher percentage of water vapour. As such, long-term hydrolytic stability is a consideration. Many different materials have been investigated for carbon capture and storage (CCS). Amine-containing solvents have been explored for some years,3 but recycling these materials on a large scale for use in power plants may require 20-40% of the energy produced. This, in turn, has been estimated by some authors to increase the cost of energy production by 70%.2 Other materials investigated as sorbents include supported amines,4-7 carbon-based sorbents,8-10 zeolites11-13 and supported carbonates.14

Microporous organic polymer networks¹⁵ combine high surface areas¹⁶ with good physicochemical stability and potential for synthetic diversification. Microporous polymer networks have been the investigated for applications such as catalysis,^{17,18} gas separation^{18,19} and gas storage,^{18,20–26} For example, the storage of hydrogen gas has been reported for a number of microporous materials^{21,22,27–31} although the volume of gas stored at room temperature is still far below the US Department of Energy targets.³² By contrast, CO₂ gas interacts more strongly with most sorbents and displays isosteric heats of sorption in the broad range required for sorption close to ambient temperature.

This journal is © The Royal Society of Chemistry 2011

CO₂ sorption properties have been reported for a range of high surface area materials³³ under varying conditions of temperature and pressure. For example, CO2 uptake in metal-organic frameworks (MOFs) is often reported at high pressures (10-50 bar)^{34–37} —that is, under conditions which are more relevant to pre-combusion capture.³³ The post-combusion capture of CO₂ might be problematic for some MOFs due their relative hydrolytic instability³⁸ although zeolitic imidazolate frameworks (ZIFs),³⁹⁻⁴¹ for example, are moisture stable and show good CO₂ uptake at atmospheric pressure. As an example, ZIF-78 has an uptake of 2.3 mmol g⁻¹ CO₂ at 298 K and 1 bar.^{39,42} Microporous organic polymers can also exhibit good physicochemical stability. It was shown recently that highly porous polyphenylene networks ($SA_{BET} > 5000 \text{ m}^2 \text{ g}^{-1}$) can be boiled in water for a week with no loss of porosity.¹⁶ There are, however, relatively a few reports of CO₂ capture in microporous polymers^{16,22,43} and those studies, too, have mainly focused on high pressure sorption conditions.

For post-combustion CO₂ capture at ambient pressure, materials with very high surface areas may not be optimal.44 Rather, increasing the heat of adsorption through the introduction of tailored binding functionalities could have more potential to increase the amount of gas adsorbed. Moreover, the ability to tune isosteric heats of sorption through chemical synthesis offers the potential to optimize the adsorption and desorption profiles of the sorbent in order to reduce energy consumption in pressureswing or temperature swing processes.⁴⁵ Recently, Torrisi et al. calculated the isosteric heats of sorption for CO₂ in functionalised MOFs and suggested that the sorption enthalpy might be modified by choice of the functional group in the organic linker.⁴⁶ They predicted that the incorporation of a carboxylic acid group would lead to the highest isosteric heat, challenging the current research emphasis in the literature regarding amine groups for CO₂ capture.^{3,47–49} The synthetic incorporation of free acid functionalities, however, might be problematic in some MOF systems due to the propensity for these groups to coordinate with metals. On the other hand, we have previously reported the

Department of Chemistry and Centre for Materials Discovery, University of Liverpool, Liverpool, United Kingdom. E-mail: aicooper@liv.ac.uk † Electronic Supplementary Information (ESI) available containing the IR spectra, TGA, pore size distributions and CO₂ adsorption isotherms See DOI: 10.1039/c1sc00100k/

synthesis of a number of functionalised conjugated microporous polymer (CMP) networks^{50–54} synthesised *via* palladium catalysed Sonogashira–Hagihara cross-coupling reactions. These networks can be synthesised with a wide range of pendant functionalities^{52,55} and with moderate to high Brunauer– Emmett–Teller (BET) surface areas. We report here the synthesis of two new networks which incorporate primary amine and carboxylic acid groups, respectively. We compare the isosteric heats and CO₂ uptakes for these polymers in addition to other functionalised CMPs and discover a surprisingly close correlation between our materials and the aforementioned computational predictions for MOFs,⁴⁶ suggesting perhaps a design principle which spans more than one class of material.

Results and discussion

Two novel functionalised conjugated polymer networks were synthesised using the Sonogashira–Hagihara palladium catalyzed cross-coupling reaction of 1,3,5-triethynylbenzene with either 2,5-dibromobenzoic acid or 2,5-dibromoaniline to yield the corresponding carboxylic acid- and amine-functionalized CMP networks, CMP-1-COOH and CMP-1-NH₂, respectively (Scheme 1).

The polymers were isolated as brown powders and analysed by FT-IR spectroscopy to show the presence of the acid and amine functionalities (Fig. S8†). A broad peak at 3421 cm⁻¹ for CMP-1-COOH shows the presence of O–H in the acid together with a strong carbonyl stretch at 1707 cm⁻¹. The N–H bonds in CMP-1-NH₂ give rise to bands at 3466 and 3376 cm⁻¹. Both networks show the presence of C=C functionalities at 2196 cm⁻¹ as well as a smaller peak at 2107 and at 3294 cm⁻¹ showing terminal alkyne groups (Fig. S8†). Further analysis of the networks by TGA showed that the networks were thermally stable to around 300 °C (see Fig. S1, Supporting Information†). The networks are also insensitive to treatment with water. Scanning electron microscopy showed that the powder morphology was similar to that reported for other CMP networks.^{55,56}

The CMP networks were analysed by nitrogen gas sorption. Adsorption/desorption isotherms (each offset for clarity by 100 cm³ g⁻¹, Fig. 1. For non-offset data, see Fig. S2†) were collected at 77 K and showed a mainly Type I shape with high gas uptake at low pressures for networks CMP-1, CMP-1-NH₂ and CMP-1-COOH with slight a slight hysteresis and Type IV isotherms with H2 hysteresis for networks CMP-1-(CH₃)₂ and CMP-1-(OH)₂ as classified by IUPAC.⁵⁷ The Brunauer–Emmett–Teller surface areas (*SA*_{BET}) were calculated from the adsorption branch of the nitrogen isotherms over a relative



Scheme 1 Synthesis of functionalised CMPs using (i) DMF, NEt₃, Pd(PPh₃)₄, CuI, 100 $^{\circ}$ C, 72 h.



Fig. 1 Nitrogen adsorption (closed)/desorption (open) isotherms (77 K) for CMP-1 (black), CMP-1-(CH₃)₂ (green), CMP-1-(OH)₂ (orange), CMP-1-NH₂ (blue) and CMP-1-COOH (red) each offset by 100 cm³ g⁻¹ for clarity.

pressure range of $P/P_0 = 0.01-0.1$ and were found to be 522 and 710 m² g⁻¹ for CMP-1-COOH and CMP-1-NH₂ respectively. Micropore volumes were calculated using the t-plot method to be 0.16 and 0.24 cm³ g⁻¹, while the total pore volumes of the two networks were 0.30 and 0.39 cm³ g⁻¹. The surface areas and pore volumes of the five networks are compared in Table 1. CMP-1-COOH and CMP-1-NH2 have the two lowest BET surface areas in this series. To estimate the contribution of microporosity to the total porosity in the networks, we have calculated the ratio of $V_{0.1}/V_{Tot}$, the ratio of the pore volume at $P/P_0 = 0.10$ to the total pore volume at $P/P_0 = 0.99$. CO₂ gas sorption measurements were also obtained for the five networks at both 273 K and 298 K. The non-functionalised CMP-1 network showed the highest volumetric CO₂ uptake at 298 K, adsorbing 1.18 mmol g^{-1} of CO₂ at 1 bar. The dihydroxy network adsorbed less CO₂ $(1.07 \text{ mmol g}^{-1} \text{ at } 1 \text{ bar})$, despite exhibiting a higher surface area and pore volume. The dimethyl network, CMP-1- $(CH_3)_2$, showed the lowest uptake of CO_2 (0.94 mmol g⁻¹), despite having a higher surface area than CMP-1. As a comparison, BPL carbon $(SA_{BET} = 1150 \text{ m}^2 \text{ g}^{-1}; \text{ a common reference material for CO}_2$ uptake) exhibits an uptake of 1.9 mmol g^{-1} . at 1 bar and 298 K.^{39,42} As an additional reference, the covalent organic framework, COF-102 ($SA_{BET} = 3620 \text{ m}^2 \text{ g}^{-1}$) shows the highest CO₂ uptake among COFs at a pressure of 55 bar and 298 K,²² however the CO₂ uptake at 1 bar and 273 K is 1.56 mmol g^{-1} -that is, less than the lowest uptake observed in this CMP series (CMP-1-COOH = 1.60 mmol g^{-1} ; see Fig S3, Supporting Information[†] for data collected at 273 K). We can therefore conclude that the CO₂ uptake in these networks at lower pressures is not dependant soley on the surface area or pore volume. It would seem that the different chemical functional groups, as well as pore size, have a large effect on the uptake of gas in this pressure-temperature regime.

The isosteric heats of adsorption were calculated from the CO_2 isotherms measured at 273 K and 298 K.^{20,26} The experimental isosteric heats showed the following order in terms of appended functional groups (Fig. 2d): –COOH > (OH)₂ > NH₂ > H > (CH₃)₂ at least for adsorbed gas quantities greater than 0.2 mmol g⁻¹. At low absorption values, CMP-1-COOH shows a heat of

Table 1 Surface areas and pore volumes for CMP networks

Network	\mathbf{R}^{a}	\mathbf{R}'^a	$SA_{\rm BET} ({ m m}^2 { m g}^{-1})^b$	$V_{\rm micro} \left({\rm cm}^3 \ {\rm g}^{-1} \right)^c$	$V_{\text{total}} \left(\text{cm}^3 \text{ g}^{-1} \right)^d$	V _{0.1/tot}
CMP-1 ^e	Н	Н	837	0.32	0.45	0.71
CMP-1-COOH	COOH	Н	522	0.22	0.30	0.73
CMP-1-NH ₂	NH_2	Н	710	0.27	0.39	0.69
$CMP-1-(CH_3)_2^e$	CH ₃	CH ₃	899	0.34	0.75	0.45
$CMP-1-(OH)_2^e$	OH	OH	1043	0.40	0.71	0.56

^{*a*} R and R' from Scheme 1. ^{*b*} Calculated over the relative pressure range P/P₀ = 0.01–0.1. ^{*c*} Pore volume at P/P₀ = 0.1. ^{*d*} Total pore volume (P/P₀ = 0.99) ^{*e*} From ref. 56

adsorption reaching 33 kJ mol⁻¹, dropping to around 27 kJ mol⁻¹ at a pressure of 1 bar — that is, substantially higher than the less polar CMP materials in this series. The variation in CO₂ sorption could potentially also arise from a difference in pore size distribution, with small pore sizes known to increase the heat of adsorption.³¹ Some differences in the pore size distribution were measured with N₂ as the probe gas (Fig. S4[†]). However, using CO₂ (which allows measurement of the smaller pores)^{58,59} the pore size distributions for all of these materials were found to be bimodal and very similar (Fig. S5[†]).

As noted above, recent computational studies for a range of functionalised MOFs based on MIL-53 predicted that functional group modification would increase the amount of CO₂ captured by the network.⁴⁶ The calculations suggested that polar groups were effective in increasing CO₂ capture, while bulky non-polar groups had a negative impact. It was predicted that the isosteric heat of adsorption would be highest for the carboxylic acid functionalised MIL, with the isosteric heats for the series to be in the order $-COOH > -(OH)_2 > -NH_2 > -(CH_3)_2 > non-functionalized. For the carboxylic acid case, the CO₂ was stabilised in$



Fig. 2 (a) Calculated CO_2 isotherms at 298 K for substituted MIL-53 frameworks, redrawn from ref. 46 and (b) measured CO_2 isotherms for CMP networks. (c) Calculated isosteric heats of adsorption for CO_2 in substituted MIL-53 frameworks, redrawn from ref. 46 and (d) measured for CMP networks. Colour-coding is as follows: unsubstituted networks (black); $-(CH_3)_2$ (green); $-(OH)_2$ (orange); $-NH_2$ (blue) and -COOH (red).

the adsorption site by the presence of interactions from neighbouring carbonyl groups. Despite the fact that our CMP networks are quite different materials (amorphous not crystalline; covalent not metal-organic), we observed essentially the same trends in isosteric heats as predicted for the MIL-53 analogues⁴⁶ (cf., Fig. 1c & 1d), thus supporting these calculations and perhaps suggesting a design principle which holds for more than one class of material. While it is not possible to fully deconvolute the effects of pore size (Fig. S4[†]) from functional group effects, our data corroborate the fundamental computational conclusion⁴⁶ that carboxylic acid functionalities are a good target for CO_2 capture materials. At this stage, we cannot say whether the adsorbed CO₂ binds cooperatively to more than one -COOH group in the CMP network, as postulated for the MIL-53 system,⁴⁶ since we do not have an unambigous structure model for these amorphous networks. It is likely however that there is a broader distribution of sorption sites in these amorphous networks, with varying propensities for multiple guest binding, and this may in part account for the broader range of isosteric heats as a function of coverage (Fig. 2d) for the experimental measurements with respect to the calculations which refer to perfectly ordered crystalline solids (Fig. 2c).46

By comparison with other materials, the heats of adsorption measured experimentally for CMP-1-COOH ($32.6-26.1 \text{ kJ} \text{ mol}^{-1}$) are 15–20% higher than those reported by Lu *et al.*⁴³ for the network PPN-1. These values are also higher than activated carbon, but lower than some MOF networks. For example, one of the highest reported heats of adsorption for CO₂ in MOFs (90 kJ mol⁻¹) occurs in the material HCu[(Cu₄Cl)₃(BTTri)₈(en)₅] where a exposed metal centre binds to an ethylene diamine ligand which in turn binds to the CO₂.⁶⁰ This binding may in fact be too strong for capture applications and might translate to a large energetic penalty. Another material with a high heat of adsorption for CO₂ is MIL-100, which contains open metal centres and exhibits a sorption enthalpy of 62 kJ mol⁻¹ at low coverage.³⁶ By contrast, the carbon material, BPL carbon, exhibits an isosteric heat of 24.3 kJ mol⁻¹.⁶¹

Conclusions

We show here that chemical composition, much more than surface area, dictates the CO_2 uptake for polymer networks at low pressures (1 atm) and close-to-ambient temperatures. The ability to fine-tune CO_2 affinity in this way is of potential value for both CCS as well as applications such as gas separation. Our results validate previous computational predictions for MOF materials,⁴⁶ suggesting that such *ab initio* computational studies have translational value within this class of materials. Our results, along with these previous calculations, also suggest that we should consider polar acidic functionalities in microporous frameworks and that such functionalities, as demonstrated here, might outperform aromatic amine functionalities for CO_2 sorption.

Acknowledgements

The authors would like to thank the Engineering and Physical Sciences Research Council (EP/G061785/1) for funding. A. I. C. is a Royal Society Wolfson Merit Award holder. The authors

gratefully acknowledge Drs A. Torrisi, R. G. Bell and C. Mellot-Draznieks at UCL for allowing the reproduction of their data (ref. 46).

Notes and references

- 1 S. Chu, Science, 2009, 325, 1599.
- 2 R. S. Haszeldine, Science, 2009, 325, 1647-1652.
- 3 G. T. Rochelle, Science, 2009, 325, 1652–1654.
- 4 X. Xu, C. Song, J. M. Andrésen, B. G. Miller and A. W. Scaroni, *Microporous Mesoporous Mater.*, 2003, 62, 29–45.
- 5 G. P. Knowles, J. V. Graham, S. W. Delaney and A. L. Chaffee, *Fuel Process. Technol.*, 2005, **86**, 1435–1448.
- 6 M. L. Gray, Y. Soong, K. J. Champagne, J. Baltrus, R. W. Stevens, P. Toochinda and S. S. C. Chuang, *Sep. Purif. Technol.*, 2004, 35, 31–36.
- 7 T. Filburn, J. J. Helble and R. A. Weiss, *Ind. Eng. Chem. Res.*, 2005, 44, 1542–1546.
- 8 M. G. Plaza, C. Pevida, A. Arenillas, F. Rubiera and J. J. Pis, *Fuel*, 2007, 86, 2204–2212.
- 9 C. Pevida, M. G. Plaza, B. Arias, J. Fermoso, F. Rubiera and J. J. Pis, *Appl. Surf. Sci.*, 2008, **254**, 7165–7172.
- 10 C. Pevida, T. C. Drage and C. E. Snape, Carbon, 2008, 46, 1464-1474.
- 11 J. Merel, M. Clausse and F. Meunier, Ind. Eng. Chem. Res., 2008, 47, 209–215.
- 12 D. Ko, R. Siriwardane and L. T. Biegler, *Ind. Eng. Chem. Res.*, 2003, **42**, 339–348.
- 13 S. Cavenati, C. A. Grande and A. E. Rodrigues, J. Chem. Eng. Data, 2004, 49, 1095–1101.
- 14 N. Shigemoto, T. Yanagihara, S. Sugiyama and H. Hayashi, *Energy Fuels*, 2006, 20, 721–726.
- 15 A. I. Cooper, Adv. Mater., 2009, 21, 1291-1295.
- 16 T. Ben, H. Ren, S. Q. Ma, D. P. Cao, J. H. Lan, X. F. Jing, W. C. Wang, J. Xu, F. Deng, J. M. Simmons, S. L. Qiu and G. S. Zhu, Angew. Chem., Int. Ed., 2009, 48, 9457–9460.
- 17 J. Schmidt, J. Weber, J. D. Epping, M. Antonietti and A. Thomas, *Adv. Mater.*, 2009, **21**, 702–705.
- 18 N. B. McKeown and P. M. Budd, Chem. Soc. Rev., 2006, 35, 675-683.
- 19 P. M. Budd and N. B. McKeown, Polym. Chem., 2010, 1, 63-68.
- 20 C. D. Wood, B. Tan, A. Trewin, H. J. Niu, D. Bradshaw, M. J. Rosseinsky, Y. Z. Khimyak, N. L. Campbell, R. Kirk, E. Stockel and A. I. Cooper, *Chem. Mater.*, 2007, **19**, 2034–2048.
- 21 P. M. Budd, A. Butler, J. Selbie, K. Mahmood, N. B. McKeown, B. Ghanem, K. Msayib, D. Book and A. Walton, *Phys. Chem. Chem. Phys.*, 2007, 9, 1802–1808.
- 22 H. Furukawa and O. M. Yaghi, J. Am. Chem. Soc., 2009, 131, 8875– 8883.
- 23 J. Germain, J. M. J. Fréchet and F. Svec, J. Mater. Chem., 2007, 17, 4989–4997.
- 24 J. Germain, J. M. J. Fréchet and F. Svec, Chem. Commun., 2009, 1526–1528.
- 25 B. S. Ghanem, K. J. Msayib, N. B. McKeown, K. D. M. Harris, Z. Pan, P. M. Budd, A. Butler, J. Selbie, D. Book and A. Walton, *Chem. Commun.*, 2007, 67–69.
- 26 C. D. Wood, B. Tan, A. Trewin, F. Su, M. J. Rosseinsky, D. Bradshaw, Y. Sun, L. Zhou and A. I. Cooper, *Adv. Mater.*, 2008, **20**, 1916–1921.
- 27 D. J. Collins and H. C. Zhou, J. Mater. Chem., 2007, 17, 3154-3160.
- 28 N. B. McKeown, B. Gahnem, K. J. Msayib, P. M. Budd, C. E. Tattershall, K. Mahmood, S. Tan, D. Book, H. W. Langmi and A. Walton, *Angew. Chem., Int. Ed.*, 2006, 45, 1804–1807.
- 29 R. E. Morris and P. S. Wheatley, Angew. Chem., Int. Ed., 2008, 47, 4966–4981.
- 30 L. J. Murray, M. Dinca and J. R. Long, Chem. Soc. Rev., 2009, 38, 1294–1314.
- 31 F. Svec, J. Germain and J. M. J. Fréchet, Small, 2009, 5, 1098-1111.
- 32 S. Satyapal, J. Petrovic, C. Read, G. Thomas and G. Ordaz, *Catal. Today*, 2007, **120**, 246–256.
- 33 D. D'Alessandro, B. Smit and J. Long, Angew. Chem., Int. Ed., 2010, 49, 6058–6082.
- 34 A. R. Millward and O. M. Yaghi, J. Am. Chem. Soc., 2005, 127, 17998–17999.
- 35 S. Couck, J. F. M. Denayer, G. V. Baron, T. Rémy, J. Gascon and F. Kapteijn, J. Am. Chem. Soc., 2009, 131, 6326–6327.

- 36 P. L. Llewellyn, S. Bourrelly, C. Serre, A. Vimont, M. Daturi, L. Hamon, G. De Weireld, J.-S. Chang, D.-Y. Hong, Y. Kyu Hwang, S. Hwa Jhung and G. R. Férey, *Langmuir*, 2008, 24, 7245– 7250.
- 37 H.-S. Choi and M. P. Suh, Angew. Chem., Int. Ed., 2009, 48, 6865– 6869.
- 38 S. S. Kaye, A. Dailly, O. M. Yaghi and J. R. Long, J. Am. Chem. Soc., 2007, 129, 14176–14177.
- 39 A. Phan, C. J. Doonan, F. J. Uribe-Romo, C. B. Knobler, M. O'Keeffe and O. M. Yaghi, *Acc. Chem. Res.*, 2010, **43**, 58–67.
- 40 R. Banerjee, A. Phan, B. Wang, C. Knobler, H. Furukawa, M. O'Keeffe and O. M. Yaghi, *Science*, 2008, 319, 939–943.
- 41 H. Hayashi, A. P. Cote, H. Furukawa, M. O'Keeffe and O. M. Yaghi, *Nat. Mater.*, 2007, 6, 501–506.
- 42 R. Banerjee, H. Furukawa, D. Britt, C. Knobler, M. O'Keeffe and O. M. Yaghi, J. Am. Chem. Soc., 2009, **131**, 3875–3877.
- 43 W. Lu, D. Yuan, D. Zhao, C. I. Schilling, O. Plietzsch, T. Muller, S. Bräse, J. Guenther, J. Blümel, R. Krishna, Z. Li and H.-C. Zhou, *Chem. Mater.*, 2010, **22**, 5964–5972.
- 44 J. R. Holst and A. I. Cooper, Adv. Mater., 2010, 22, 5212-5216.
- 45 D. Aaron and C. Tsouris, Sep. Sci. Technol., 2005, 40, 321-348.
- 46 A. Torrisi, R. G. Bell and C. Mellot-Draznieks, *Cryst. Growth Des.*, 2010, **10**, 2839–2841.
- 47 N. Hiyoshi, K. Yogo and T. Yashima, *Microporous Mesoporous Mater.*, 2005, 84, 357–365.
- 48 V. Zelenak, D. Halamova, L. Gaberova, E. Bloch and P. Llewellyn, *Microporous Mesoporous Mater.*, 2008, 116, 358–364.
- 49 J. C. Hicks, J. H. Drese, D. J. Fauth, M. L. Gray, G. Qi and C. W. Jones, J. Am. Chem. Soc., 2008, 130, 2902–2903.

- 50 J. X. Jiang, F. Su, A. Trewin, C. D. Wood, N. L. Campbell, H. Niu, C. Dickinson, A. Y. Ganin, M. J. Rosseinsky, Y. Z. Khimyak and A. I. Cooper, *Angew. Chem., Int. Ed.*, 2007, 46, 8574–8578.
- 51 J.-X. Jiang, F. Su, A. Trewin, C. D. Wood, H. Niu, J. T. A. Jones, Y. Z. Khimyak and A. I. Cooper, J. Am. Chem. Soc., 2008, 130, 7710–7720.
- 52 J.-X. Jiang, A. Trewin, F. Su, C. D. Wood, H. Niu, J. T. A. Jones, Y.Z. Khimyak and A. I. Cooper, *Macromolecules*, 2009, **42**, 2658–2666.
- 53 E. Stöckel, X. F. Wu, A. Trewin, C. D. Wood, R. Clowes, N. L. Campbell, J. T. A. Jones, Y. Z. Khimyak, D. J. Adams and A. I. Cooper, *Chem. Commun.*, 2009, 212–214.
- 54 J.-X. Jiang, A. Laybourn, R. Clowes, Y. Z. Khimyak, J. Bacsa, S. J. Higgins, D. J. Adams and A. I. Cooper, *Macromolecules*, 2010, 43, 7577–7582.
- 55 R. Dawson, A. Laybourn, R. Clowes, Y. Z. Khimyak, D. J. Adams and A. I. Cooper, *Macromolecules*, 2009, 42, 8809–8816.
- 56 R. Dawson, A. Laybourn, Y. Z. Khimyak, D. J. Adams and A. I. Cooper, *Macromolecules*, 2010, 43, 8524–8530.
- 57 K. S. W. Sing, D. H. Everett, R. A. W. Haul, L. Moscou, R. A. Pierotti, J. Rouquerol and T. Siemieniewska, *Pure Appl. Chem.*, 1985, 57, 603–619.
- 58 D. Lozano-Castelló, D. Cazorla-Amorós and A. Linares-Solano, Carbon, 2004, 42, 1233–1242.
- 59 J. Weber, J. Schmidt, A. Thomas and W. Böhlmann, *Langmuir*, 2010, 26, 15650–15656.
- 60 A. Demessence, D. M. D'Alessandro, M. L. Foo and J. R. Long, J. Am. Chem. Soc., 2009, 131, 8784–8786.
- 61 K. B. Lee, M. G. Beaver, H. S. Caram and S. Sircar, *Ind. Eng. Chem. Res.*, 2008, 47, 8048–8062.